

# The importance of the cathode specific interfacial area in modeling a cylindrical alkaline cell

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## Abstract

Low initial cell voltage predictions from a mathematical model of a Zn/MnO<sub>2</sub> cylindrical alkaline cell are corrected by increasing the cathode specific interfacial area. A simplified model of the cathode region is used to estimate the initial polarization loss when the cathode specific interfacial area is changed, for a particular AA-size design and galvanostatic discharge rate. The model expression for the cathode specific interfacial area is reformulated to match the literature values for this quantity more closely. A new formulation is accepted as the revision, and the discharge time improvement is substantial with respect to the base case when simulations using the complete model are performed. The physical effects of raising the cathode specific interfacial area are analyzed. The increased discharge time leads to a localized region in the cathode that is totally depleted of zincate ion. This creates numerical problems in the cell voltage calculation. © 1999 Elsevier Science S.A. All rights reserved.

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## 1. Introduction

The mathematical model presented by Podlaha and Cheh [1,2] for a cylindrical Zn/MnO<sub>2</sub> alkaline cell has been used extensively in computer simulation of the AA-size configuration. For example, the model was applied in two studies that optimized the cell behavior by adjusting the initial cathode porosity [3] and altering the active material loading in both electrodes [4]. These four studies indicate that the model gives realistic predictions of the AA-size cell behavior. Another observation is that the model is responsive to changes in certain parameters that are considered influential with respect to the cell performance.

However, there remain certain physical discrepancies that need to be addressed prior to additional modeling of the Zn/MnO<sub>2</sub> system. In battery research, the cell voltage profiles are the most important results that are generated from a first principles model. In the case of the Zn/MnO<sub>2</sub> system, the predicted initial cell voltage is considered too low in comparison to actual cells. This tendency of the

model to underpredict the operating voltage, because of a low initial cell voltage, is especially problematic for the high-rate discharge regime. In this case, a large cell current causes large activation overpotentials, so that the calculated discharge time to reach the cell cutoff voltage may be quite inaccurate because of the low value of the initial cell voltage.

The cathode region of the Zn/MnO<sub>2</sub> system limits the cell performance [5] in that most of the polarization losses in discharging are produced in this region because of the charge transfer and solid-state diffusion resistances. Thus, any method to correct the low predictions of the initial cell voltage should start with an analysis of the polarization losses associated with the cathode region. There are two approaches available to raise the initial cell voltage predictions [6–8]. The first technique is to vary nonphysical parameters in the governing equations of the model in an attempt to find arbitrary parameter values that raise the initial cell voltage. The second procedure is to find a rational argument for the revision of certain parameters that have a physical basis in the model, and that are related to the cell polarization.

A simplified model is developed to describe the electrochemical reaction rate in the cathode region, which pro-

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vides for an estimation of the initial polarization loss in the cell voltage calculation. This simplified model depends largely on the single parameter,  $a_c$ , the specific interfacial area in the cathode region, under the assumption that the exchange current density evaluated at a reference condition for the cathode electrochemical reaction is invariant. Studies that focus on a small number of important parameters in a battery system are available in the literature [9,10]. The cathode specific interfacial area is a measurable, physical quantity. The results of a literature search are presented to demonstrate that the cathode specific interfacial area is too low in the complete model. A method is proposed to revise the model expression for  $a_c$  that uses the simplified description of the cathode as a guide, and simulations using the complete model are performed to identify an appropriate form of  $a_c$ . Comparisons of modeling predictions are made between simulation results using the current and revised  $a_c$  forms, so that an understanding of the physical effects of changing  $a_c$  can be obtained. The entire analysis is conducted under the assumption that a published base case AA-size configuration [1–4] is the test design being considered.

## 2. Simplified cathode model

A detailed overview of the Zn/MnO<sub>2</sub> cylindrical alkaline cell is listed elsewhere [1]. Only a brief system description is provided here. The primary Zn/MnO<sub>2</sub> cylindrical alkaline cell initially contains a porous Zn anode and a porous MnO<sub>2</sub> cathode, with a concentrated KOH aqueous electrolyte. When discharged, Zn is oxidized to form zincate ion, Zn(OH)<sub>4</sub><sup>2-</sup>, which precipitates to form solid zinc oxide, ZnO, by homogeneous chemical reaction in the anode and separator regions. In the cathode, MnO<sub>2</sub> is reduced to form manganese oxyhydroxide, MnOOH, during the first-electron discharge of MnO<sub>2</sub>.

It is assumed that the polarization losses in the anode region due to electrochemical reaction and ohmic effects, as well as the voltage losses in the separator region because of ohmic effects, are negligible with respect to the activation polarization losses in the cathode region [5]. The important reaction is then the single-electron reduction of MnO<sub>2</sub>, shown as



The complete cell model [1] expresses the transfer current form of this reaction as a modified Butler–Volmer equation, with additional terms that incorporate the solid-state proton diffusion. The proton diffusion term reflects the assumption of a mixed activation and diffusion control regime.

In the simplified model, the voltage loss of the entire cell is assumed to occur only because of electrochemical reaction in the cathode. The solid-state diffusion resistance

in the cathode is not considered. Furthermore, the porous cathode is considered a pseudo-planar electrode, so that the transfer current is taken as a function of the cell current with no spatial dependence. For large values of  $\eta_{c,sm}$ , the cathode overpotential in the context of the simplified model, a Tafel form of the transfer current is used. The result is

$$j_{c,sm} = -a_c i_0 \exp\left(-\frac{\alpha_c F}{RT} \eta_{c,sm}\right) \quad (2)$$

Here,  $j_{c,sm}$  is the transfer current in the cathode using the simplified model,  $a_c$  is the cathode specific interfacial area,  $i_0$  is the exchange current density for the cathode reaction evaluated at a reference condition,  $\alpha_c$  is the cathodic transfer coefficient for the cathode reaction, and  $T$  is the cell temperature. For small values of the cathode overpotential, a linear transfer current expression is appropriate, with the form

$$j_{c,sm} = a_c i_0 \frac{(\alpha_a + \alpha_c) F}{RT} \eta_{c,sm} \quad (3)$$

where  $\alpha_a$  is the anodic transfer coefficient for the cathode reaction.

Since  $i_0$  is taken as a constant,  $j_{c,sm}$ ,  $a_c$ , and  $\eta_{c,sm}$  can then be considered time-dependent quantities that do not depend on position. The transfer current is approximated by

$$j_{c,sm} = -\frac{I}{V_{\text{cath}}} \quad (4)$$

where  $I$  is the cell current and  $V_{\text{cath}}$  is the cathode volume. For a galvanostatic discharge,  $j_{c,sm}$  is constant. Under this scenario of a particular constant cell current value, the relationship between  $a_c$  and  $\eta_{c,sm}$  becomes apparent. A larger value of  $a_c$  will yield a value for  $\eta_{c,sm}$  that is lower in absolute magnitude, and vice versa. Therefore, from the perspective of the simplified model of the cathode, if  $a_c$  is too low the result may be a prediction of  $\eta_{c,sm}$  that is too large in absolute magnitude. Thus, an inspection of the model value for  $a_c$  is justified.

## 3. Cathode specific interfacial area

### 3.1. Formulation for $a_c$ in the complete model

The model computes the cathode specific interfacial area through the assumption of identical, spherical solid reactant particles with a time-dependent outer radius denoted by  $r_0$  [11], which increases throughout a discharge [1]. The cathode specific interfacial area is calculated from

$$a_c = 4\pi N(r_0)^2 \quad (5)$$

where  $N$  is the number density of  $\text{MnO}_2$  particles in the cathode volume. It is seen that the only recourse to literature values occurs through  $r_0$ . Thus, the model presently does not consider the many sources of values for the specific interfacial area of electrolytic manganese dioxide (EMD), the most common battery grade form of  $\text{MnO}_2$ . The  $\text{Zn}/\text{MnO}_2$  model assumes that the  $\text{MnO}_2$  active material in the cathode is EMD [1].

Table 1 lists values for  $N$  and  $r_0^0$ , the initial value of the  $\text{MnO}_2$  particle radius, for the base case AA-size design. This yields an initial cathode specific interfacial area,  $a_c^0$ , of  $1200 \text{ cm}^{-1}$ . The above expression for  $a_c$  is not readily observed in the battery literature. In the complete model formulation,  $a_c$  is a function of position in the cathode region and time.

### 3.2. Literature values and the relation to the $\text{Zn}/\text{MnO}_2$ model

A literature search is performed in order to evaluate the accuracy of the model in determining  $a_c$ . Let  $a_{\text{EMD}}^0$  denote the literature values reported for EMD with the units of  $\text{m}^2/\text{g-MnO}_2$ , which indicates the specific interfacial area available at the start of discharging. The search results show that a high-end estimate of  $a_{\text{EMD}}^0$  is on the order of  $100 \text{ m}^2/\text{g-MnO}_2$  [13–24].

These values are not on a per unit electrode volume basis and must be adjusted by using

$$a_c^0 \times 10^{-4} = a_{\text{EMD}}^0 \rho_{\text{EMD}} \epsilon_{\text{EMD}}^0 \quad (6)$$

where  $\rho_{\text{EMD}}$  is the true EMD density,  $\epsilon_{\text{EMD}}^0$  is the initial volume fraction of EMD in the cathode, and the numerical factor on the left hand side is a conversion factor, because the model requires that  $a_c$  have units of  $\text{cm}^{-1}$ , for the

particle surface area per unit cathode volume. The initial volume fraction of EMD,  $\epsilon_{\text{EMD}}^0$ , is related to the initial cathode porosity,  $\epsilon_c^0$ , through

$$\epsilon_{\text{EMD}}^0 = 1 - \epsilon_c^0 - \epsilon_G^0 \quad (7)$$

where  $\epsilon_G^0$  is the initial volume fraction of graphite, which is assumed to be 10 wt.% of the total solid material in the cathode [3,4]. The values for  $\rho_{\text{EMD}}$ ,  $\epsilon_{\text{EMD}}^0$ ,  $\epsilon_c^0$  and  $\epsilon_G^0$  for the base case design are found in Table 1.

The above expression for the cathode specific interfacial area gives a high-end estimate of  $a_c^0$  on the order of  $10^6 \text{ cm}^{-1}$ . This is three orders of magnitude larger than the current model value. The current value for  $a_c^0$  is thus considered too low. Therefore, with reference to the simplified cathode model, the low value for  $a_c^0$  may be responsible for the low initial cell voltage predicted by the complete model.

### 3.3. Formulation to revise $a_c$

The simplest way to revise  $a_c$  is to slightly modify the original expression in the model. This is achieved by adding a numerical factor,  $k_c$ , in the calculation of  $a_c$ . The result is

$$a_c = k_c 4\pi N (r_0)^2 \quad (8)$$

where  $k_c$  is dimensionless. This expression appears to be an arbitrary adjustment of  $a_c$ , but the literature values for  $a_{\text{EMD}}^0$  serve as a guide to find more realistic  $a_c$  values. This formulation to revise  $a_c$  preserves the size-change dynamics of the solid particles in the cathode.

Zhang and Cheh [25] have used an effective discharge factor in their correlations of constant load experimental discharges in order to fit the cell voltage data most accurately. The effective discharge factor itself is a pre-factor to the transfer current expression in the cathode and serves as a de facto adjustment of  $a_c$ . However, literature values for  $a_c$  when the discharge factor was used, as well as a physical analysis of the performance effect due to the discharge factor, were not included in the study. In contrast, a goal of this study is to examine the physical effects of revising  $a_c$  in the model when attempting to match the literature values of  $a_{\text{EMD}}^0$  more closely.

## 4. Initial cell voltage

### 4.1. Simplified model

The simplified model in the cathode region is now used to estimate the initial cell voltage. This is performed by calculating the initial voltage loss in the cathode region.

Table 1  
System quantities

$i_0$ cathode	$2.0 \times 10^{-7} \text{ A/cm}^2$ [1]
$\alpha_a$ cathode	0.5 [1]
$\alpha_c$ cathode	0.5 [1]
$T^0$	298.15 K [1]
$I$	1.0 A [1]
$V_{\text{cath}}$	$3.07 \text{ cm}^3$ [4]
$j_{c,\text{sm}}$	$-0.326 \text{ A/cm}^3$
$N$	$2.400 \times 10^7 \text{ cm}^{-3}$ [11]
$r_0^0$	$20.0 \times 10^{-4} \text{ cm}$ [1]
$\rho_{\text{EMD}}$	$4.370 \text{ g/cm}^3$ [4]
$\epsilon_{\text{EMD}}^0$	0.626 [4]
$\epsilon_c^0$	0.24 [1]
$\epsilon_G^0$	0.134 [4]
$E_{\text{ocp}}(0)$	1.6 V [12]
$r_s$	0.45 cm [1]
$r_c$	0.66 cm [1]
$\kappa^0$	$0.6 \Omega^{-1} \text{ cm}^{-1}$ [3]
$\sigma_c$	$19.8 \Omega^{-1} \text{ cm}^{-1}$ [1]
Initial KOH concentration	$0.007 \text{ mol/cm}^3$ [1]
Initial $\text{K}_2\text{Zn}(\text{OH})_4$ concentration	$5.3 \times 10^{-4} \text{ mol/cm}^3$ [1]

For the Tafel regime, Eqs. (2) and (4) are combined to give

$$\eta_{c,sm}(0) = -\frac{RT^0}{\alpha_c F} \ln\left(\frac{I}{a_c^0 i_0 V_{cath}}\right) \quad (9)$$

Similarly, for the linear case, Eq. (3) is rearranged to give

$$\eta_{c,sm}(0) = -\frac{RT^0}{(\alpha_a + \alpha_c) F} \frac{I}{a_c^0 i_0 V_{cath}}. \quad (10)$$

In both of these equations,  $\eta_{c,sm}(0)$  refers to the initial overpotential value in the cathode, and  $I$  is considered constant. The values needed to compute the initial cathode overpotential for both cases are listed in Table 1, which also gives the value of  $j_{c,sm}$  for the base case design. The initial cell voltage is then calculated by

$$E_{sm}(0) = E_{ocp}(0) + \eta_{c,sm}(0) \quad (11)$$

where  $E_{sm}(0)$  is the initial cell voltage calculated from the simplified model, and  $E_{ocp}(0)$  is the initial open circuit potential of the cell, which is given in Table 1.

#### 4.2. Full model

The complete Zn/MnO<sub>2</sub> mathematical model is referred to as the full model or complete model. The initial cell voltage calculated by the full model considers all of the relevant polarization losses at the beginning of discharge, which include contributions from the anode and separator regions, in addition to the cathode region. The initial cell voltage calculated by the full model is expressed by

$$E_{fm}(0) = E_{ocp}(0) - \eta_a(0) + \eta_s(0) + \eta_{c,fm}(0) \quad (12)$$

where  $\eta_a(0)$  and  $\eta_s(0)$  are the anode and separator total initial overpotentials, and  $\eta_{c,fm}(0)$  is the total initial cathode overpotential for the full model. By varying  $k_c$  it is possible to perform simulations using the full model so that  $\eta_{c,sm}(0)$  and  $\eta_{c,fm}(0)$  can be compared. This also leads to the comparison of  $E_{sm}(0)$  and  $E_{fm}(0)$ .

### 5. Secondary current distribution

When the geometrical configuration of the base case AA-size design is fixed, the current distribution in the cathode is governed by three parameters [11,26,27]. They are a dimensionless cell current,  $\delta$ , a dimensionless exchange current density,  $\nu^2$ , and the ratio of the matrix to electrolyte effective conductivities,  $\gamma$ . Since the only parameter change that is studied here involves an increase in  $a_c$ , only  $\nu^2$  will be altered. The values of  $\delta$  and  $\gamma$  for the base case design are 3.997 and  $2.807 \times 10^2$ , respectively [3]. These two numbers indicate that the reaction rate in

the cathode is nonuniform and skewed towards the separator/cathode interface.

The expression for  $\nu^2$  is

$$\nu^2 = \frac{(\alpha_a + \alpha_c) F a_c^0 i_0 (r_c - r_s)^2}{RT^0} \left( \frac{1}{\kappa^0 (\epsilon_c^0)^{1.5}} + \frac{1}{\sigma_c} \right) \quad (13)$$

where  $r_s$  is the anode/separator interface location,  $r_c$  is the cathode current collector location,  $\kappa^0$  is a characteristic value of the electrolyte conductivity, and  $\sigma_c$  is the cathode effective matrix conductivity [3]. The values needed to calculate  $\nu^2$ , except for  $a_c^0$ , are given in Table 1. Since a larger value is sought for  $a_c^0$ ,  $\nu^2$  will increase with respect to its base case value. Increasing  $\nu^2$  will have the effect of creating a more nonuniform current distribution in the cathode.

## 6. Results

### 6.1. Overview

Simulations using the complete model are performed with different values of  $a_c^0$ , through different  $k_c$  values, using the same base case AA-size design published in earlier studies [1–4]. The operating condition is a 1.0 A discharge with a 0.8 V cutoff voltage. The numerical solution of the model has been described previously [1]. Five different cases are studied and they are summarized in Table 2. The base case design is given by  $k_c = 1$ . The discharge time is increased substantially as  $k_c$  is increased. The depth of discharge at the cell cutoff voltage,  $f_d$ , is also shown for each case. Greater utilization of cathode active material occurs as the discharge time is increased, due to the larger  $a_c$  values. The initial cell voltage, as predicted by the simplified and complete models, increases significantly as  $a_c$  increases.

When the initial voltage losses for the simplified model and the complete model are compared, the results are quite similar. The same is true for the initial cell voltages calculated by the full model and the simplified model. The desire to increase the initial cell voltage by raising  $a_c$  is thus validated by the simplified model, and more importantly by the complete model. The major result of the simplified model is the degree to which the initial cathodic voltage losses computed by the simplified model,  $\eta_{c,sm}(0)$ , match those of the complete model,  $\eta_{c,fm}(0)$ . These values are extremely close and appear to differ by about 50 mV for each  $a_c^0$  value.

The simplified model underpredicts the initial cathodic voltage loss computed by the full model. For each  $a_c^0$  value, the combined initial overpotential from the anode

Table 2  
Simplified model results and simulation results from the full model

Quantity	$k_c = 1$	$k_c = 10$	$k_c = 50$	$k_c = 100$	$k_c = 1000$
$a_c^0, \text{cm}^{-1}$	1200	$1.20 \times 10^4$	$6.03 \times 10^4$	$1.20 \times 10^5$	$1.20 \times 10^6$
Regime	Tafel	Tafel	Tafel	Tafel	Linear
$\eta_{c,\text{sm}}(0), \text{V}$	-0.370	-0.252	-0.169	-0.134	-0.035
$E_{\text{sm}}(0), \text{V}$	1.230	1.348	1.431	1.466	1.565
$\eta_{c,\text{fm}}(0), \text{V}$	-0.425	-0.307	-0.224	-0.189	-0.084
$E_{\text{fm}}(0), \text{V}$	1.163	1.281	1.363	1.399	1.504
$\nu^2$	$5.919 \times 10^{-3}$	$5.919 \times 10^{-2}$	0.2960	0.5919	5.919
$t_d, \text{h}$	0.700	1.006	1.150	1.189	1.278
$f_d$	0.277	0.395	0.451	0.466	0.500

and separator regions, as calculated by the full model, is nearly constant and is approximately 10 mV. This suggests that the initial polarization loss due to ohmic effects in the cathode accounts for the rest of the deviation between the initial cathodic overpotential values for the simplified and complete models. This additional polarization is significant, although the simplified model gives results that are surprisingly close to the full model. Therefore, the simplified model of the cathode is deemed a good approximation for the purposes of this study.

A new formulation for  $a_c$  is sought for the complete model, using a single  $k_c$  value. This is the simplest way to implement the revision. Although Zhang and Cheh [25] varied the effective discharge factor for different constant load values, the selection of a single value of  $k_c$  for the model revision is justified by the assumption of a reliable value of  $a_{\text{EMD}}^0$  in the cell loadings, such that  $a_c$  requires an upward adjustment. However, the actual  $a_{\text{EMD}}^0$  values that are encountered in practice are not available, so a low-end value of  $k_c$  is chosen to revise the  $a_c$  expression. The choice of a low-end  $k_c$  value ensures that the cell performance will not be overpredicted. The revised  $a_c$  expression is therefore chosen with  $k_c = 50$ , corresponding to  $a_c^0 = 6.03 \times 10^4 \text{ cm}^{-1}$ . This value is in accordance with the effective discharge factor values listed by Zhang and Cheh. The rest of this study will compare predictions made by the full model for the base case and revised  $a_c$  expressions, or for the  $k_c = 1$  and  $k_c = 50$  simulations, respectively.

### 6.2. Cell voltage

Fig. 1 shows the cell voltage profiles for the five  $k_c$  values detailed in Table 2. The initial cell voltage is increased as  $k_c$  is raised, and the discharge time is substantially lengthened when  $k_c$  rises in value. It is observed that the  $k_c = 50$  case captures the main adjustment in discharge time, meaning that above  $k_c = 50$ , the increase in cell life diminishes.

Above  $k_c = 10$ , voltage fluctuations appear as the cutoff voltage is approached. These occur because the longer cell life leads to the situation where a localized region of zero zincate ion concentration is encountered. Physically, this

represents an interesting conceptual problem where the electrolyte reverts to a binary system in an isolated region. It is analogous to a problem treated by Hauser and Newman [28], but their results are not applicable in this study. The voltage fluctuations are considered to represent a numerical discrepancy to be addressed in future work. The zincate ion behavior is discussed in a later section.

The base case simulation with  $k_c = 1$  is inferior to the other cell voltage profiles. The  $k_c = 50$  case gives a 64% increase in the discharge time relative to the base case. This is a major improvement in simulated performance, and it is the result of the revision of a single, physically relevant parameter.

### 6.3. Transfer current

Fig. 2 shows transfer current profiles for the base case with  $k_c = 1$ , and Fig. 3 displays transfer current profiles for the revised case with  $k_c = 50$ . The profiles are shown for the cathode region. Table 2 lists the  $\nu^2$  values as  $k_c$  is raised. For  $k_c = 50$ ,  $\nu^2$  is increased by about two orders of magnitude with respect to the base case. The current distribution is expected to be more nonuniform for the  $k_c = 50$  profiles. Figs. 2 and 3 support this prediction.

The base case transfer current profiles are nonuniform, and they are large in magnitude at the separator/cathode interface. The greater nonuniformities are seen in Fig. 3 for  $k_c = 50$ , especially at longer times. A large reaction front penetrates the cathode to a larger extent than with the base case. Also, the reaction rate at the separator/cathode interface is lowered significantly as the discharge progresses. However, the reaction still does not penetrate the cathode near the cathode current collector, which is supported by the large ratio of matrix to electrolyte effective conductivities that is unchanged by the revision of  $a_c$ . The simplified model value of  $j_{c,\text{sm}} = -0.326 \text{ A/cm}^3$  is a decent characteristic value for the transfer current in both figures.

### 6.4. Local overpotential

Figs. 4 and 5 detail the local overpotential profiles in the cathode for  $k_c = 1$  and  $k_c = 50$ , respectively. It is

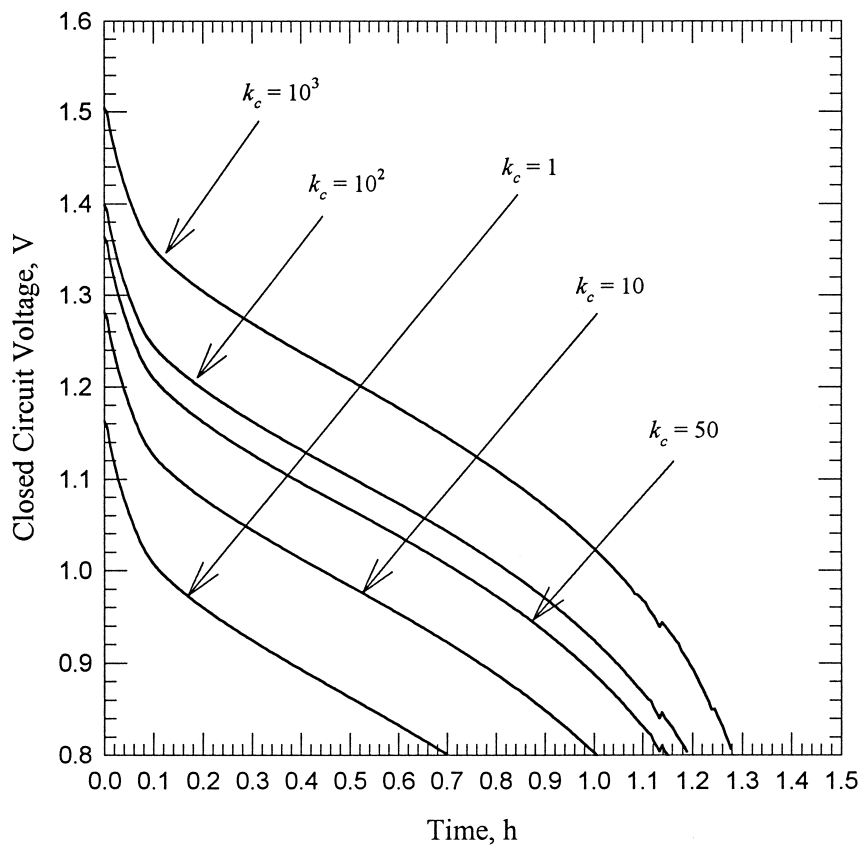


Fig. 1. Cell voltage curves with different expressions for the cathode specific interfacial area.

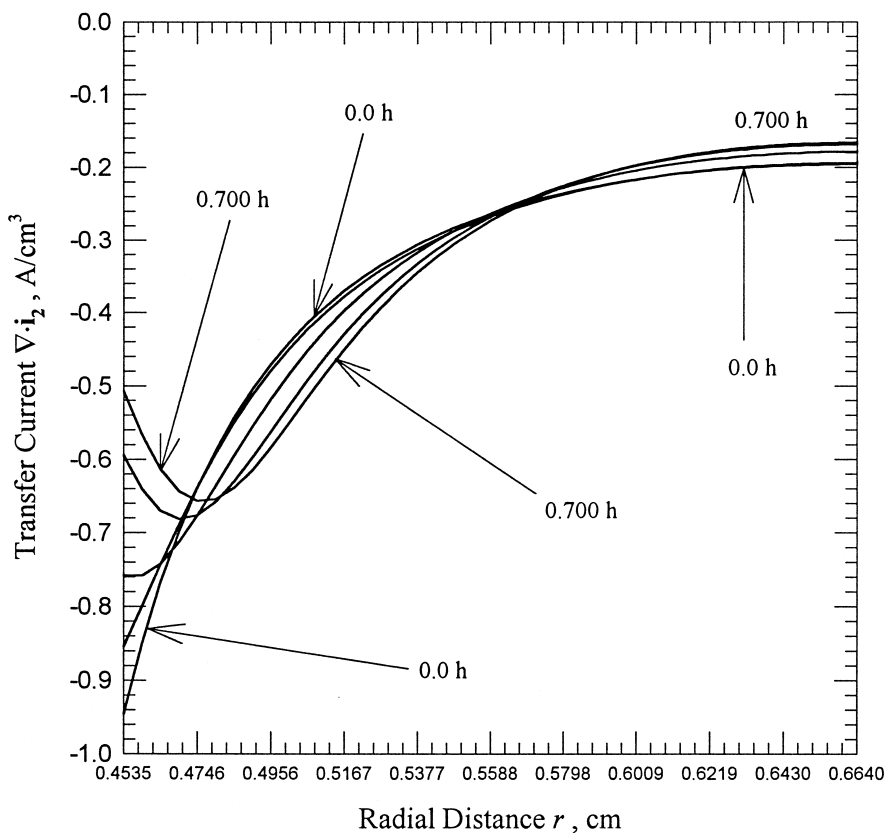


Fig. 2. Transfer current profiles throughout the cathode for the base case cathode specific interfacial area expression, with  $k_c = 1$ .

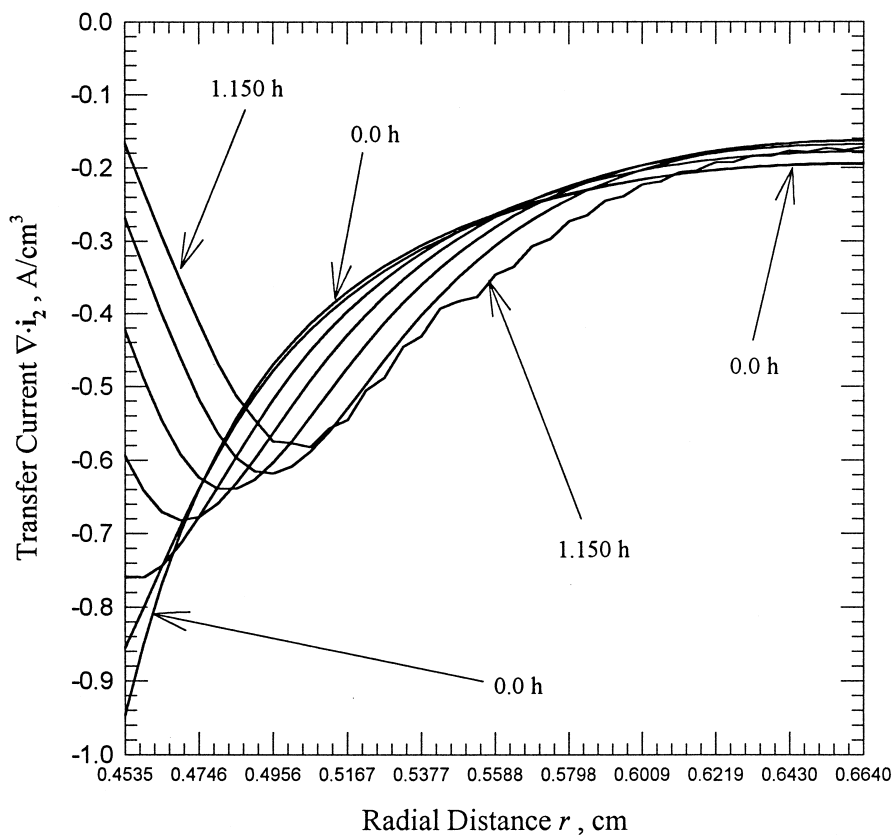


Fig. 3. Transfer current profiles throughout the cathode for the revised cathode specific interfacial area expression, with  $k_c = 50$ .

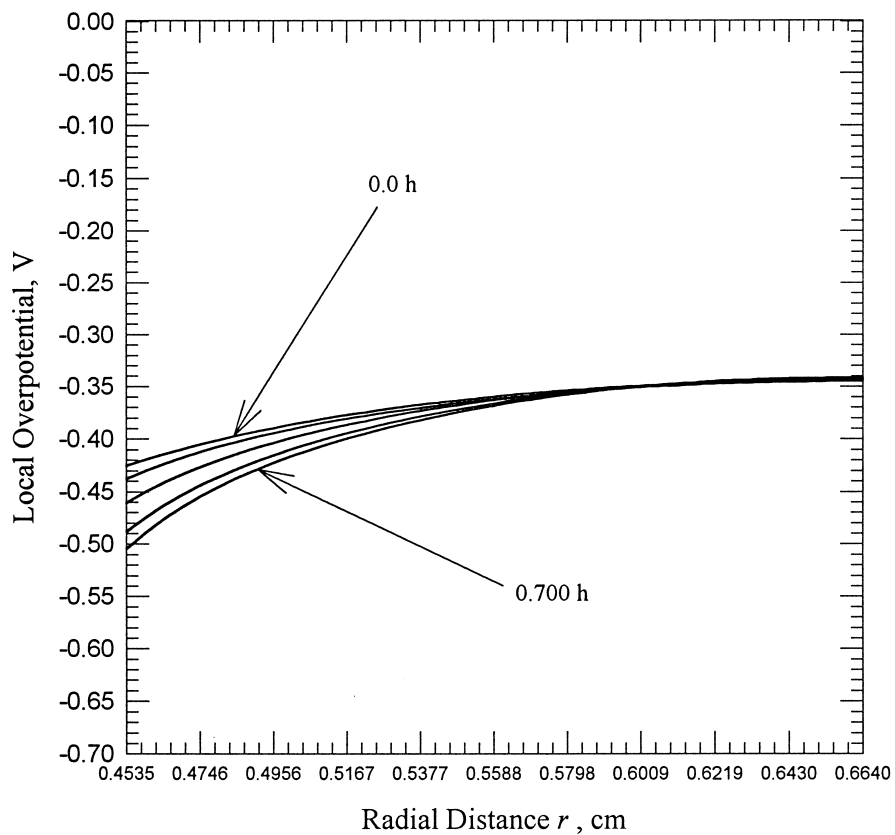


Fig. 4. Local overpotential profiles throughout the cathode for the base case cathode specific interfacial area expression, with  $k_c = 1$ .

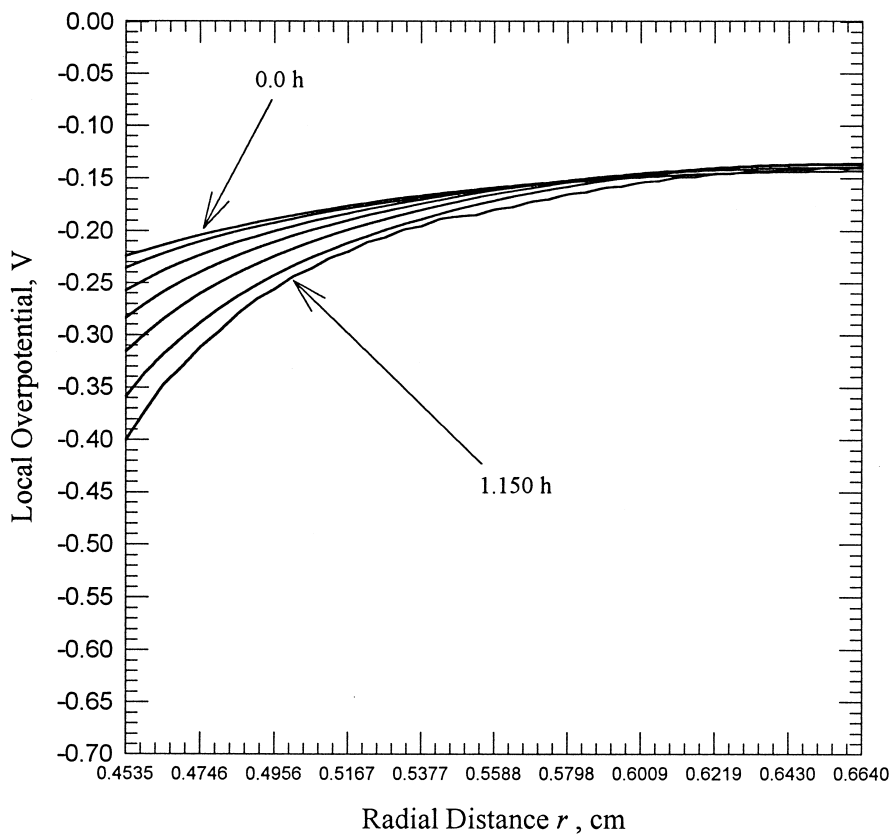


Fig. 5. Local overpotential profiles throughout the cathode for the revised cathode specific interfacial area expression, with  $k_c = 50$ .

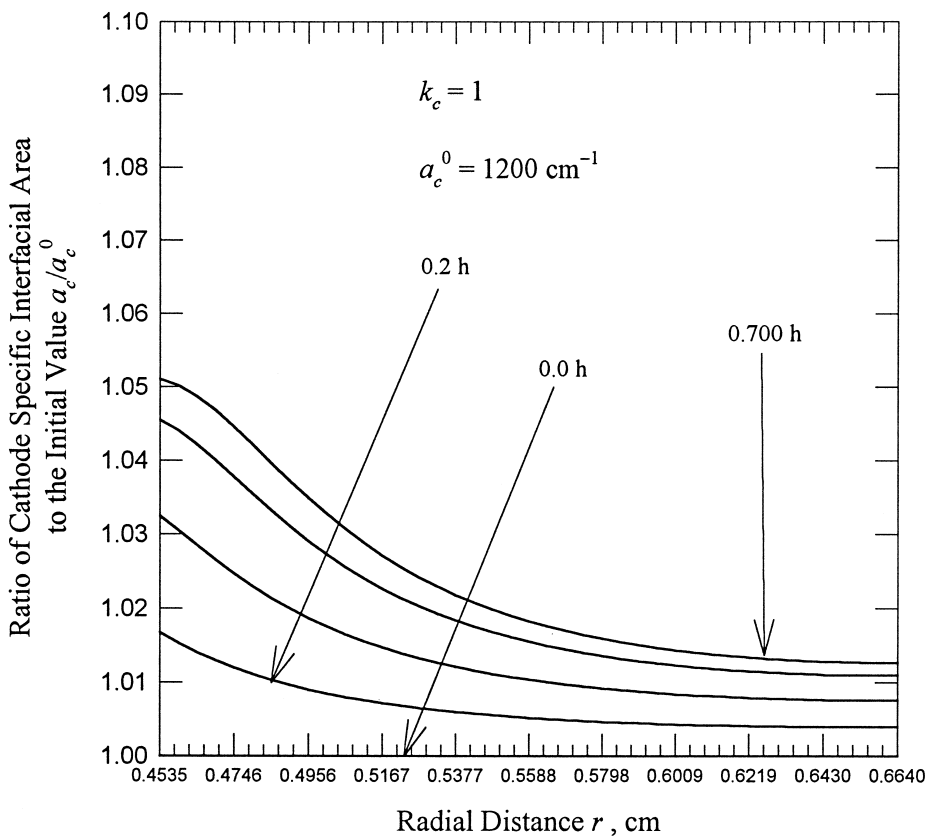


Fig. 6. Cathode specific interfacial area profiles for the base case  $a_c$  expression, with  $k_c = 1$ .



easily seen that the local overpotential is greatly lowered in magnitude when  $a_c$  is increased. The cell voltage calculation in the complete model depends on the local overpotential evaluated at three positions within the cell [3], where the cathode delivers the main contribution to the cell voltage loss. Thus, the lower overpotential values in the cathode directly explain the higher operating voltage, and the much longer cell life, when  $a_c$  is increased. Therefore, the appearance of lower overpotentials in the cathode is the major effect of revising  $a_c$ . A key relationship that is predicted by the secondary current distribution is seen, whereby a larger  $a_c$  leads to lower local overpotential profiles, but the reaction rate profiles become less uniform.

### 6.5. Cathode specific interfacial area

The profiles for  $a_c$  in the cathode for  $k_c = 1$  and  $k_c = 50$  are shown, respectively, in Figs. 6 and 7. The specific interfacial area increases with time in both cases. This is because  $r_0$  increases with time [1], and the effect of  $r_0$  is seen in Eq. (8). For the base case, the increase in  $a_c$  is limited to about 5%. The largest change occurs near the separator/cathode interface because of the larger reaction rate. The revised form of  $a_c$  with  $k_c = 50$  shows similar behavior, although the more nonuniform reaction rate in

the cathode leads to a slightly larger increase in  $a_c$ , which achieves a maximum of about 7% at a short distance away from the separator/cathode interface. The changes in  $a_c$  throughout a discharge are not drastic, and an argument can be made for using a constant value of  $a_c$  in the complete model.

A possible flaw in the model is the use of  $r_0$  in the calculation of  $a_c$ . This is because  $r_0$  describes the outer surface of a reacted solid particle that contains MnOOH in the outer shell and MnO<sub>2</sub> in the inner core, with the result of an overpredicted value of  $a_c$ . It may be more appropriate to use the inner radius that describes the MnO<sub>2</sub> shell only, which decreases throughout a discharge. However, this radius does not change very much throughout a discharge, as is the case with  $r_0$ , so that the effect of using  $r_0$  may not be important in relation to the revised  $a_c$  formulation with the new  $k_c$  value. This also suggests that a constant  $a_c$  value may be satisfactory.

### 6.6. Zincate ion concentration

Figs. 8 and 9 present the zincate ion concentration profiles throughout the entire cell for the base case and revised  $a_c$  expressions, or with  $k_c = 1$  and  $k_c = 50$ , respectively. In these figures the anode is on the left side, the separator is between the two vertical, dotted lines, and the

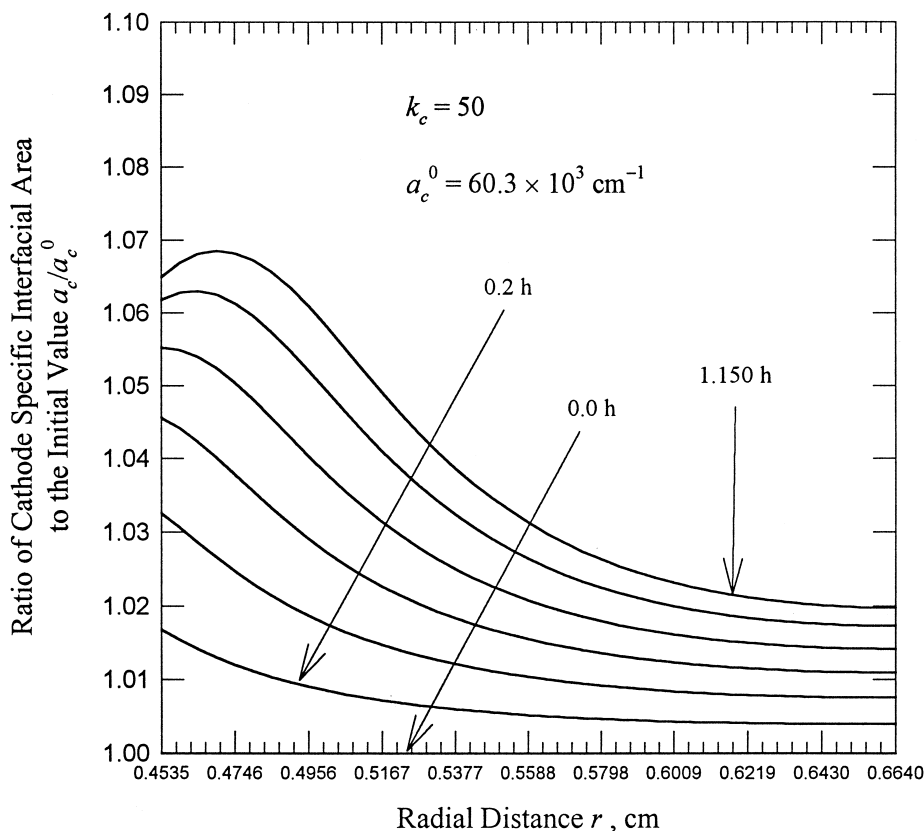


Fig. 7. Cathode specific interfacial area profiles for the revised  $a_c$  expression, with  $k_c = 50$ .

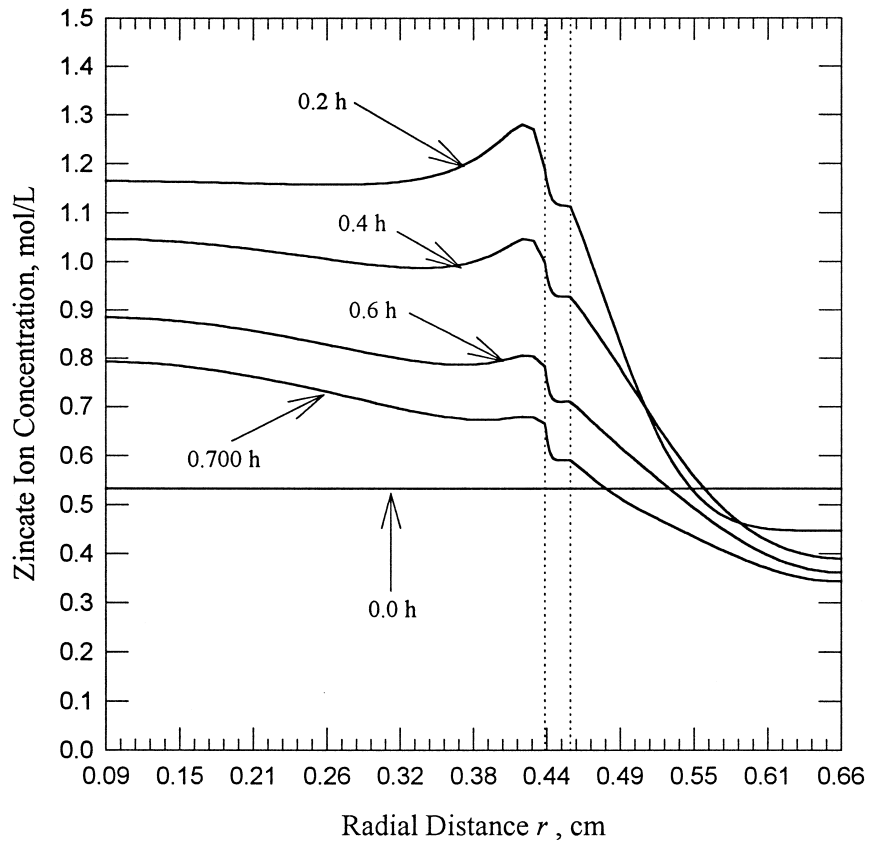


Fig. 8. Zincate ion concentration profiles throughout the entire cell for the base case cathode specific interfacial area expression, with  $k_c = 1$ .

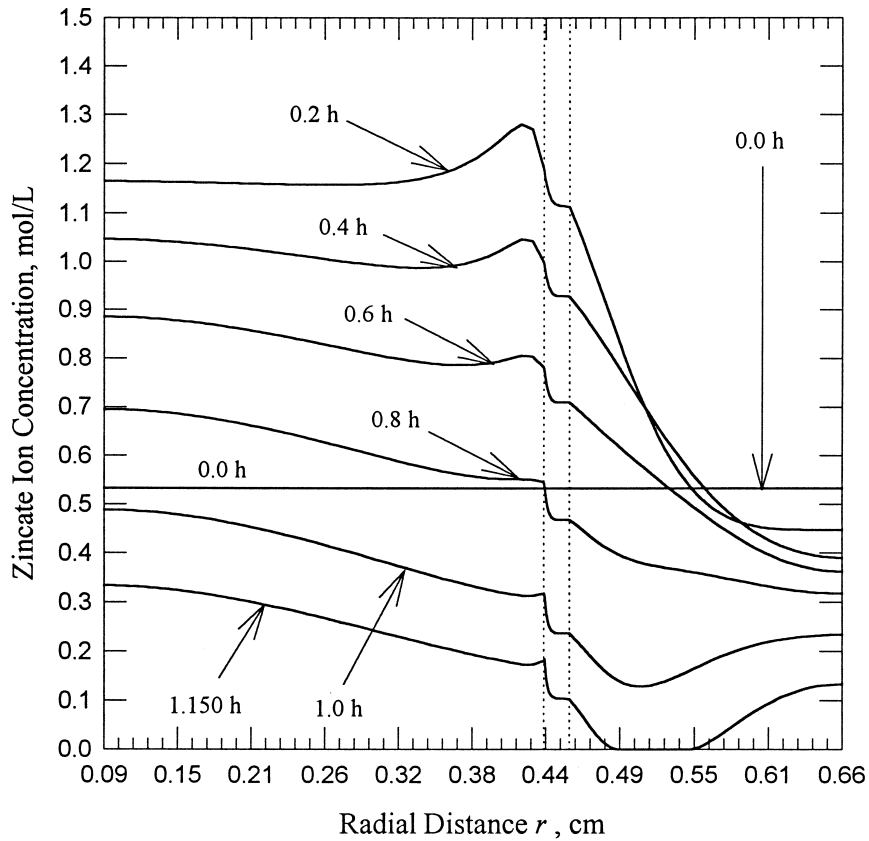


Fig. 9. Zincate ion concentration profiles throughout the entire cell for the revised cathode specific interfacial area expression, with  $k_c = 50$ .

cathode is on the right side. The initial KOH and potassium zincate ( $K_2Zn(OH)_4$ ) concentrations are given in Table 1. At short times immediately after discharging begins, there is a large accumulation of zincate ion in the electrolyte. The profiles start to decrease as the discharge time increases. The profiles for both cases are nearly identical throughout the discharge time for the base case. But the much longer discharge time for the revised  $a_c$  case leads to the depletion of zincate ion in a confined region in the cathode. This is the cause of the voltage fluctuations that appear in Fig. 1.

When the zincate ion concentration is zero, the solution becomes a binary electrolyte in a localized region, yet remains a ternary electrolyte throughout the rest of the cell. The form of Ohm's law in the electrolyte phase changes when the ternary to binary electrolyte transition occurs [1]. Because of the logarithmic dependence on concentration, it is difficult to numerically implement the ternary to binary electrolyte transition without the adverse fluctuations in voltage. It is believed that the actual cause of zincate ion depletion observed here is due to the large increase in discharge time when  $a_c$  is revised, which precludes consideration of some other type of explanation. The cathode specific interfacial area does not have a direct effect on the zincate ion behavior. However, flux of zincate ion out of the cathode occurs because of migrational forces at longer times [1]. The voltage fluctuations with the revised  $a_c$  expression are unacceptable numerical inconsistencies that are to be corrected in another study.

## 7. Conclusions

Low initial cell voltage predictions are corrected by revising the formulation for  $a_c$ . The revision of the cathode specific interfacial area demonstrates the major effect that a single, physical parameter can have in the predictions of the Zn/MnO<sub>2</sub> mathematical model. A low-end adjustment to the formulation of  $a_c$ , which is supported by the literature, leads to a large simulated increase in cell life. The revised form of  $a_c$  is taken with  $k_c = 50$ , which gives  $a_c^0 = 6.03 \times 10^4 \text{ cm}^{-1}$ . A simplified mathematical model of the cathode region as an idealized, spatially invariant electrode provides extremely accurate estimations of the initial polarization loss experienced by the AA-size test design as  $a_c$  is adjusted, relative to predictions made by the complete mathematical model. The revised  $a_c$  expression leads to the prediction of a more nonuniform reaction rate in the cathode, and a higher operating voltage is observed because of local overpotential profiles that are reduced in magnitude in the cathode. Finally, the revised  $a_c$  formulation leads to the depletion of zincate ion in a localized region in the cathode. This creates numerical

fluctuations in the cell voltage calculation that must be corrected.

## 8. List of symbols

$a_c$	Specific interfacial area in the cathode, defined as the solid particle surface area per unit cathode volume, $\text{cm}^{-1}$
$a_{\text{EMD}}$	Specific interfacial area of electrolytic manganese dioxide as reported in the literature, $\text{m}^2/\text{g-MnO}_2$
$E_{\text{ocp}}(0)$	Initial open circuit potential of the cell, V
$E_{\text{fm}}(0)$	Initial cell voltage calculated by the full model, V
$E_{\text{sm}}(0)$	Initial cell voltage calculated by the simplified model, V
$f_d$	Depth of discharge at the cell cutoff voltage
$F$	Faraday's constant, 96,487 C/mol
$i_0$	Exchange current density for the cathode reaction evaluated at a reference condition, $\text{A}/\text{cm}^2$
$I$	Cell current, A
$j_{c,\text{sm}}$	Transfer current in the cathode region as calculated by the simplified model, $\text{A}/\text{cm}^3$
$k_c$	Factor used in Eq. (8) for the modified cathode specific interfacial area expression
$N$	Number density of MnO <sub>2</sub> particles in the cathode region available for electrochemical reaction, $\text{cm}^{-3}$
$r_0$	Outer radius of a solid reactant particle contained in the cathode, cm
$r_c$	Cathode current collector location, cm
$r_s$	Separator/cathode interface location, cm
$R$	Universal gas constant, 8.3143 J/mol K
$t_d$	Discharge time to reach the cutoff voltage, h
$T$	Cell temperature, K
$V_{\text{cath}}$	Cathode volume, $\text{cm}^3$
<i>Greek</i>	
$\alpha_a$	Anodic transfer coefficient for the cathode electrochemical reaction
$\alpha_c$	Cathodic transfer coefficient for the cathode electrochemical reaction
$\gamma$	Ratio of matrix to electrolyte effective electrical conductivities in the cathode
$\delta$	Dimensionless cell current using cathode parameters
$\epsilon_i$	Porosity of region $i$ , or solid volume fraction of species $i$
$\eta_a$	Total overpotential in the anode region, V
$\eta_s$	Total overpotential in the separator region, V
$\eta_{c,\text{fm}}$	Total overpotential in the cathode region as calculated by the full model, V

$\eta_{c,sm}$	Total overpotential in the cathode region as calculated by the simplified model, V
$\kappa$	Electrolyte conductivity, $\Omega^{-1} \text{ cm}^{-1}$
$\nu$	Square root of the dimensionless exchange current density in the cathode
$\rho_i$	Density of component $i$ , $\text{g}/\text{cm}^3$
$\sigma_c$	Cathode effective matrix conductivity, $\Omega^{-1} \text{ cm}^{-1}$

*Superscripts*

0	Initial condition or characteristic value
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